

DOE HLW Glass Degradation Model



Material Recovery & Waste Form Development

William Ebert Argonne National Laboratory

NWTRB Briefing Washington D.C. May 15, 2017



NWTRB questions addressed

Nuclear Energy

May 15, 2017 pre-meeting briefing questions for DOE:

- Provide an overview of the compositions and projected quantities of existing and future HLW glass at the West Valley Demonstration Project, Savannah River Site, and the Hanford Site (including "German" glass logs).
 - How is the variability in DOE HLW glass composition taken account of in DOE's glass corrosion models?
 - How well are the glass corrosion model parameters supported by experimental data?
- What is the status of DOE R&D activities to understand and model the long-term performance of borosilicate HLW glass?
 - Which R&D activities are run or managed by the different DOE offices and programs [DOE-NE (including NEUP), DOE-EM, DOE Office of Science (if any)] and how are these activities integrated? What are the accomplishments?
 - A detailed plan for joint DOE-NE and DOE-EM R&D activities on glass corrosion initially was developed in 2011 (Ryan et al. 2011)1 that included experiments and modeling. What are the status and results of the tasks described in the plan?
 - How are the results of international R&D activities integrated with the results of DOE R&D?
- From DOE's perspective, what are the remaining technical uncertainties and gaps in data and understanding of the long-term performance of HLW glass? How is DOE addressing these uncertainties and gaps?
- How is DOE integrating process-level models of HLW glass corrosion and radionuclide release into generic repository performance assessments?
 - How is the DOE approach to HLW glass performance modeling different from that for the low-activity waste (LAW) glass to be disposed of at the Hanford Site Integrated Disposal Facility?
 - What lessons learned from LAW glass corrosion experiments and modeling can be applied to HLW glass?

• What is the technical basis for extrapolating the results of short-term, small-scale tests on glass corrosion to long-term glass waste form performance in a repository?



Overview

- ANL developed a bounding glass dissolution model to represent all US HLW glasses ca. 2002
- Used well-established mechanistically-based rate expression for borosilicate glass dissolution to limit radionuclide release rates
 - Based on mineral dissolution kinetics (e.g., Lasaga 1983, Aagaard and Helgeson 1982)
 - Modified for application to borosilicate waste glasses (e.g., Grambow 1985)
- Extracted dependencies on glass composition, pH, and temperature from QA-compliant test data
 - Confirmed general application of dependencies by comparisons with literature data
- Represented dependencies on solution composition probabilistically by using bounding values
 - Confirmed applicability of model to wide range of waste glass compositions by comparisons with literature data



BACKGROUND: Application to Glass (based on B. Grambow 1985)

Nuclear Energy

$$rate_{G} = k_{0} \times 10^{\eta \cdot pH} \times \left[\exp\left(\frac{-E_{a}}{RT}\right) \right] \times \left[1 - \frac{Q}{K} \right] + k_{long}$$

- $rate_G$ = glass dissolution rate (mass area⁻¹ time⁻¹) η = empirical pH dependence (unitless) Q = ion activity product (molar) R = gas constant (kJ mol⁻¹ K⁻¹) k_{long} = minimum glass dissolution rate (mass area⁻¹ time⁻¹)
- k_0 = intrinsic glass dissolution rate (mass area⁻¹ time⁻¹) E_a = effective activation energy (kJ mol⁻¹) K = effective glass solubility product (molar) T = temperature (K)
- The rate-limiting step for glass corrosion is reaction of the -OSi(OH)₃ end member

 $\equiv \text{Si-O-Si(OH)}_3 + \text{H}_2\text{O} \rightarrow \equiv \text{Si-OH} + \text{H}_4\text{SiO}_4$

- The saturation index defining the reaction affinity is the ratio of the activity of orthosilicic acid $[Q = a(H_4SiO_4)]$ and the stability constant for the above reaction (K).
- The constant term k_{long} was included in the rate expression to prevent the calculated rate from becoming zero if the value of Q became equal to (or greater than) K in simulations over long durations (Grambow 1985).



Quantify Dependence of Rate on pH and Temperature

$$rate_{G} = k_{0} \times 10^{\eta \cdot pH} \times \left[\exp\left(\frac{-E_{a}}{RT}\right) \right] \times \left[1 - \frac{Q}{K} \right] + k_{long}$$

- Determined values for temperature dependence parameter E_a and pH dependence parameter η likely to be bounding for HLW glasses
 - Used data from short-term ASTM C1220 tests in which value of (1-Q/K) remains near 1 and k_{long} is negligible, so that

$$rate_G = k_0 \times 10^{\eta \cdot pH} \times \left[\exp\left(\frac{-E_a}{RT}\right) \right]$$

- Extracted parameter values of η and E_a based on boron release of from series of ASTM C1220 tests conducted at constant pH and temperature
- Compared with dependencies extracted from data in literature



ASTM C1220 (MCC-1) Method

Nuclear Energy

Monolith specimen: 10 cm^3 leachant/cm² area (S/V = 10 m^{-1})

steel reaction vessel at 90 °C

Batch tests conducted for different durations in DIW buffered at different pH

Analyzed solution for dissolved glass constituents



Measured concentrations were normalized to S/V ratio and mass fraction of element in each glass:

$$NL(i) = \frac{[i]}{(S/V)f(i)}$$

Normalized mass loss NL(i) has units g glass m⁻².



Temperature and pH Coefficients Determined from ASTM C1220 Tests with SRL 202G Glass

Nuclear Energy

e.g., 10 m⁻'; pH 5.0; 90 ℃



log rate = log (1.27) = 0.104

Dissolution rates were determined graphically as $\Delta NL(B)/\Delta t$ in units g glass m⁻²d⁻¹.

Roll over is due to decreasing affinity term and only solid points were used to determine rate.

pH values were measured at room temperature.





Comparison with Values Derived from Literature Data

Glass	Temperature	η	E _a (k.l/mol)	Reference				
Acidic Solutions								
CSG	25, 50, 70	-0.70	60	Knauss et al. 1990				
MW	30, 50, 70, 90	-0.43	32	Abraitis et al. 2000				
Binder Glass	40, 70, 90	-0.36	72	Fanning et al. 2003				
DOE model		-0.49	31					
	Alkaline Solutions							
CSG	40, 70, 90	0.49	85	Knauss et al. 1990 ^a				
MW	30, 50, 70, 90	0.43	56	Abraitis et al. 2000				
LD6-5412	20, 40, 70, 90	0.40	75	McGrail, et al. 1997				
R7T7	90	—	59	Delage and Dussossoy 1991				
R7T7	90	0.39		Gin et al. 1994				
Binder Glass	40, 70, 90	0.64	83	Fanning et al. 2003				
DOE model		0.49	69					



Quantify Effect of Glass Composition

- Determined parameter value k₀ for glass composition dependence that is likely to be bounding for all HLW glasses
 - Used data from tests in which value of (1-Q/K) remains near 1 and k_{long} is negligible, so that

$$rate_{G} = k_{0} \times 10^{\eta \cdot pH} \times \left[\exp\left(\frac{-E_{a}}{RT}\right) \right]$$

- Conducted ASTM C1220 (MCC-1) tests with nine reference glasses to measure glass composition dependence in alkaline solutions
- Deconvolved pH and temperature terms from measured rate to calculate k_0
- Determined how much k_0 varied with glass composition



Nuclear Energy

Glass Composition Dependence k_0 Determined from ASTM C1220 Tests at 90 °C with Several Glasses

							rate		
	0.000	Elemental Mass %				NR(B),	- Ll	log ₁₀ (k ₀)	
	Glass	AI	В	Fe	Na	Si	g/(m²⋅d)	рп	[g/(m ² ·d)]
	LD6-5412	6.82	1.66	0.09	15.0	27.5	0.47	9.3	5.04
	Hanford-L	6.33	2.75	4.03	14.8	17.9	0.97	9.5	5.26
HLW -	Hanford-D	5.36	2.17	16.1	11.7	14.1	1.8	10.5	5.04
	WV ref 6'	3.17	4.00	8.41	5.93	19.2	0.69	9.5	5.11
	SRL 51S	2.79	2.30	8.53	7.11	26.3	0.66	9.9	4.90
	SRL 165U	2.16	2.10	8.21	8.04	24.7	1.0	9.6	5.23
	SRL 202U	2.03	2.48	7.97	6.61	22.9	0.69	9.8	4.97
	SRL 131U	1.73	3.00	8.85	8.95	20.4	1.2	9.8	5.21
	PNL 76-68	0	2.79	6.41	10.5	19.8	1.1	9.2	5.46

 $rate_G = k_0 \cdot 10^{\eta pH} \cdot exp(-E_a/RT)$

for all 9 glasses: $log(k_0) = 5.14 \pm 0.17$ (1 σ) relative standard deviation = 3.3% log(k₀) = 5.07 \pm 0.13 (1 σ) relative standard deviation = 2.6%

Effect of glass composition is taken into account adequately by pH and T terms.



Quantify Effect of Solution Composition

$$rate_{G} = k_{0} \times 10^{\eta \cdot pH} \times \left[\exp\left(\frac{-E_{a}}{RT}\right) \right] \times \left[1 - \frac{Q}{K}\right] + k_{long}$$

- Value of Q (which is [H₄SiO₄]) changes over time as glass, EBS materials, etc. dissolve, alteration phase precipitate, and ground water flows.
- Value of (1-Q/K) and glass dissolution rate change as value of Q changes:
 - For borosilicate waste glasses: $Q = a(H_4SiO_4)$ $K_{glass} = constant$
 - Glass dissolution rate slows as solution approaches saturation
- Secondary phase formation affects glass dissolution rate



Experimental Observation of Stage 3

Nuclear Energy



Increases in B, Na, and Si releases are attributed to the generation of secondary phases such as analcime at Point P that affect reaction affinity for glass dissolution.



Reaction Progress Plot





Bounding Glass Model

- Conditions triggering/controlling Stage 3 rate not fully understood
- Analytical form of affinity term for Stage 3 not known
- Bounding model approach used at that time
- Used lumped term to bound effects of glass composition and reaction affinity
- Potential changes in glass dissolution rate due for Stage 3 processes are bounded in the model by using a probability distribution for rate in alkaline solutions that provides
 - upper rate that bounds possible Stage 3 rates
 - lower rate that bounds residual rates
- Advances in understanding and modeling Stage 3 behavior will be discussed in later talks



Modeling Effects of Solution Composition

Nuclear Energy

$$rate_{G} = k_{0} \times 10^{\eta \cdot pH} \times \left[\exp\left(\frac{-E_{a}}{RT}\right) \right] \times \left[1 - \frac{Q}{K}\right] + k_{long}$$

Combine dependency on solution composition with glass composition term in one lumped term $k_{\rm E} = k_0 \cdot (1 - Q/K)$

acidic rate =
$$k_{E, acid} \cdot 10^{-0.49pH} \cdot exp(-31/RT)$$

alkaline rate = $k_{E, alkaline} \cdot 10^{0.49pH} \cdot exp(-69/RT)$

- Maximum and minimum values of k_E were used to bound dissolution rates measured for representative HLW glasses under different test conditions
 - Maximum model rates provide upper bound to measured Stage 3 rates
 - Minimum model rates represent or provide lower bound for k_{long}



Bounding k_E Values

- Minimum k_E in acidic solutions = 8.41 x 10³ g m⁻² d⁻¹
 - based on rates measured in ANL drip tests with a representative DWPF glass made with SRL 165 frit. (Fortner and Bates 1996)
 - Expected to be most probable for acidic solutions.
- Minimum k_E in alkaline solutions = 2.82 × 10¹ g m⁻² d⁻¹
 - derived from rates in vapor hydration tests with representative DWPF and WVDP glasses. (Ebert 2003b)
 - Expected to be most probable for alkaline solutions.
- Maximum k_E in acidic solutions = 1.15 x 10⁷ g m⁻² d⁻¹
 - derived from rates in C1220 tests with a representative DWPF glass SRL 202G. (Ebert 2003a)
- Maximum k_E in alkaline solutions = 3.47 × 10⁴ g m⁻² d⁻¹
 - based on rates from 7-day PCT results for set of representative DWPF, WVDP, and Hanford glasses. (Ebert 2003a)



Example: Values of log k_E Derived from Results of 7-d PCT at 90 °C

Nuclear Energy

Glass	NL(B), g/m ²	NR(B), g/(m²·d)	pH(room temp)	log k _F , g/(m ² ·d)
SRL 51S a	0.267	0.0381	10.66	3.29 3 36
SRL 51S b	0.247	0.0353	10.33	3.42 5
SRL 165U	0.308	0.0440	10.31	3.52
SRL 202U	0.298	0.0426	10.42	3.45
SRL 131U	4.81	0.687	11.63	4.07
SRL202G	0.608	0.0869	11.11	3.42
WV ref 6	0.270	0.0386	9.98	3.63
Hanford-D	0.361	0.0516	10.67	3.41
PNL 76-68	1.23	0.171	9.43	4.54
Hanford-L	0.475	0.0679	10.96	3.39
LD6-5412	0.082	0.0117	11.20	2.51
EA	8.21	1.17	11.87	4.18
				3.57 ± 0.97 (2σ)
		F /	、 ¬	

$$\log_{10} k_{E} = \log_{10} NR(B) - 0.49 \times pH - \log_{10} \left[exp\left(\frac{-69 \text{ kJ/mol}}{RT}\right) \right]$$

 $mean + 2\sigma$ $log_{10} k_{E} = 4.54$ $k_{E} = 3.47 \times 10^{4} \text{ g m}^{-2} \text{ d}^{-1}$ 17



Example: Stage 3 Rates Measured at 90 °C and Derived log k_E Values

Nuclear Energy

					DOE model
Glass	NR(B), g/(m².d)	рН	log k _E , g/(m²⋅d)	Reference	Max. rate g/(m²·d)
EA	0.070	12.3	2.75	Ebert et al. 1998	4.35
SRL 131A	0.037	12.1	2.57	Ebert and Bates 1993	3.47
SRL 202A	0.032	12.0	2.55	Ebert and Bates 1993	3.10
SRL 200S	0.87	12.2	3.89	Feng et al. 1993	3.88
SAN60	0.074	9.8	4.00	Patyn et al. 1990	0.09
LD6-5412	0.40	12.0	3.65	Ebert, Bakel, and Brown 1996	1.04

Stage 3 rates are bounded by using the maximum rates calculated with $\log_{10} k_E = 4.54$

DOE glass model ca. 2004

$$rate_{G} = \mathbf{k}_{E} \cdot 10^{\eta pH} \cdot \exp(-\mathbf{E}_{a}/R\mathbf{T})$$
Glass model parameters: $\mathbf{k}_{E}, \eta, \mathbf{E}_{a}$ Environmental parameters: \mathbf{pH}, \mathbf{T}



Maximum and Minimum Model Glass Dissolution Rates in Alkaline Solutions (Illustrative)





NWTRB questions addressed

Nuclear Energy

How is the variability in DOE HLW glass composition taken account of in DOE's glass corrosion models?

The effect of glass composition on the corrosion rate is taken into account using an intrinsic rate coefficient, which was determined to be only weakly affected by glass composition. In the bounding model, the glass composition term is lumped with the reaction affinity term and the range of values used for the lumped parameter selected to bound measured dissolution rates.

- How well are the glass corrosion model parameters supported by experimental data?

The glass corrosion model parameter values were derived from appropriate experimental data and shown to be consistent with results available in the literature.

• What is the technical basis for extrapolating the results of short-term, small-scale tests on glass corrosion to long-term glass waste form performance in a repository?

The results of short-term, small-scale tests are not extrapolated to represent long-term performance in DOE models. Rather, models are based on technically defensible rate equations and laboratory tests are used to determine model parameter values that bound long-term performance in disposal facilities.



References

- Aagaard, P. and Helgeson, H.C. 1982. "Thermodynamic and Kinetic Constraints on Reaction Rates Among Minerals and Aqueous Solutions. I. TheoreticalConsiderations." American Journal of Science, 282, (3), 237-285. New Haven, Connecticut: Yale University.
- Abraitis, P.K.; Livens, F.R.; Monteith, J.E.; Small, J.S.; Trivedi, D.P.; Vaughan, D.J.; and Wogelius, R.A. 2000. "The Kinetics and Mechanisms of Simulated British Magnox Waste Glass Dissolution as a Function of pH, Silicic Acid Activity, and Time in Low Temperature Aqueous Systems." Applied Geochemistry, 15, (9), 1399-1416. New York, New York: Pergamon.
- Delage, F. and Dussossoy, J.L. 1991. "R7T7 Glass Initial Dissolution Rate Measurements Using a High-Temperature Soxhlet Device." Scientific Basis for Nuclear Waste Management XIV, Symposium held November 26-29, 1990, Boston, Massachusetts. Abrajano, T., Jr. and Johnson, L.H., eds. 212, 41-47. Pittsburgh, Pennsylvania: Materials Research Society.
- Ebert, W.L., and Jerden, J.L. Jr. (2016). Implementation of the ANL Stage 3 Glass Dissolution Model. DOE NE report FCRD-MRWFD-2016-000296.
- Ebert, W.L. 2003a. "Data Report for MCC-1 Tests and PCTs with SRL 202G Glass." Memo from W.L. Ebert to File, July 24, 2003, with attachment. ACC: MOL.20030728.0214; MOL.20030728.0215.
- Ebert, W.L. 2003b. "Vapor Hydration Tests with DWPF Reference Glasses." Memo from W.L. Ebert to File, July 24, 2003, with attachment. ACC: MOL.20030728.0211; MOL.20030728.0212.
- Ebert, W.L.; Buck, E.C.; Luo, J.S.; Tam, S.W.; and Bates, J.K. 1998. Corrosion Behavior of Environmental Assessment Glass in Product Consistency Tests of Extended Duration. ANL-98/27. Argonne, Illinois: Argonne National Laboratory.
- Ebert, W.L.; Bakel, A.J.; and Brown, N.R. 1996. "Measurement of the Glass Dissolution Rate in the Presence of Alteration Phases." Proceedings of the International Topical Meeting on Nuclear and Hazardous Waste Management, Spectrum '96, August 18-23, 1996, Seattle, Washington. Pages 569-575. La Grange Park, Illinois: American Nuclear Society.
- Ebert, W.L. and Bates, J.K. 1993. "A Comparison of Glass Reaction at High and Low Glass Surface/Solution Volume." Nuclear Technology, 104, 372-384. Hinsdale, Illinois: American Nuclear Society.
- Fanning, T.H.; Ebert, W.L.; Frank, S.M.; Hash, M.C.; Morris, E.E.; Morss, L.R.; O'Holleran, T.P.; and Wigeland, R.A. 2003. Status of Ceramic Waste Form Degradation and Radionuclide Release Modeling. ANL-03/8. Argonne, Illinois: Argonne National Laboratory.



References cont.

- Feng, X.; Bates, J.K.; Buck, E.C.; Bradley, C.R.; and Gong, M. 1993. "Long-Term Comparison of Dissolution Behavior Between Fully Radioactive and Simulated Nuclear Waste Glasses." *Nuclear Technology*, *104*, 193-206. Chicago, Illinois: American Nuclear Society.
- Fortner, J.A. and Bates, J.K. 1996. "Long-Term Results from Unsaturated Durability Testing of Actinide-Doped DWPF and WVDP Waste Glasses." Scientific Basis for Nuclear Waste Management XIX, Symposium held November 27-December 1, 1995, Boston, Massachusetts. Murphy, W.M. and Knecht, D.A., eds. 412, 205-211.
- Pittsburgh, Pennsylvania: Materials Research Society. TIC: 233877.Gin, S.; Godon, N.; Mestre, J.P.; Vernaz, E.Y.; and Beaufort, D. 1994. "Experimental Investigation of Aqueous Corrosion of R7T7 Nuclear Glass at 90 °C in the Presence of Organic Species." *Applied Geochemistry*, *9*, (3), 255-269. New York, New York: Pergamon.
- Grambow, B. 1985. "A General Rate Equation for Nuclear Waste Glass Corrosion. "Scientific Basis for Nuclear Waste Management VIII, Symposium held November 26-29, 1984, Boston, Massachusetts. Jantzen, C.M.; Stone, J.A.; and Ewing, R.C., eds. 44, 15-27. Pittsburgh, Pennsylvania: Materials Research Society.
- Jégou, C.; Gin, S.; and Larché, F. 2000. "Alteration Kinetics of a Simplified Nuclear Glass in an Aqueous Medium: Effects of Solution Chemistry and of Protective Gel Properties on Diminishing the Alteration Rate." *Journal of Nuclear Materials*, 280, (2), 216-229. New York, New York: North-Holland.
- Knauss, K.G.; Bourcier, W.L.; McKeegan, K.D.; Merzbacher, C.I.; Nguyen, S.N.; Ryerson, F.J.; Smith, D.K.; Weed, H.C.; and Newton, L. 1990. "Dissolution Kinetics of a Simple Analogue Nuclear Waste Glass as a Function of pH, Time and Temperature." *Scientific Basis for Nuclear Waste Management XIII, Symposium held November 27-30, 1989, Boston, Massachusetts.* Oversby, V.M. and Brown, P.W., eds. 176, 371-381. Pittsburgh, Pennsylvania: Materials Research Society.
- Lasaga, A.C. 1983. "Rate Laws of Chemical Reactions." Chapter 1 of *Kinetics of Geochemical Processes*. Lasaga, A.C. and Kirkpatrick, R.J., eds. Reviews in Mineralogy Volume 8. Washington, D.C.: Mineralogical Society of America.
- McGrail, B.P.; Ebert, W.L.; Bakel, A.J.; and Peeler, D.K. 1997. "Measurement of Kinetic Rate Law Parameters on a Na-Ca-Al Borosilicate Glass for Low-Activity Waste." *Journal of Nuclear Materials*, 249, 175-189. Amsterdam,
- Patyn, J.; Van Iseghem, P.; and Timmermans, W. 1990. "The Long-Term Corrosion and Modeling of Two Simulated Belgian Reference High-Level Waste Glasses-Part II." *Scientific Basis for Nuclear Waste Management XIII, Symposium held November 27-30, 1989, Boston, Massachusetts*. Oversby, V.M. and Brown, P.W., eds. 176, 299-307.
 Pittsburgh, Pennsylvania: Materials Research Society.



Acknowledgements

Nuclear Energy

This work is supported by the U.S. Department of Energy, Office of Nuclear Energy, Materials Recovery and Waste Form Development Campaign work package FT-17-AN030105.

Government License Notice: this presentation was created by UChicago Argonne, LLC, Operator of Argonne National Laboratory ("Argonne"). Argonne, a U.S. Department of Energy Office of Science laboratory, is operated under Contract No. DE-AC02-06CH11357. The U.S. Government retains for itself, and others acting on its behalf, a paid-up nonexclusive, irrevocable worldwide license in said presentation to reproduce, prepare derivative works, distribute copies to the public, and perform publicly and display publicly, by or on behalf of the Government.